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Characterization of Nanocomposites SnO₂:Cu for Gas Sensing Applications

The paper discusses the importance of improving and developing gas sensors, with a special focus on tin oxide nanoparticles doped with Cu nanoparticles for NO₂ detection. The structural, morphological and optical properties of the SnO₂ nanofilms were revealed using XRD, FE-SEM and UV-Vis characterization. The structure of the material was attributed to the tetragonal crystalline configurations of SnO₂. With the enhancement of Cu doping, the absorbance decreased with an increase in the optical bandgap values from 3.02 to 3.28 eV. An increase in the sensitivity values to nitrogen dioxide gas was reported for the films prepared from pure SnO₂ and incorporated with Cu nanoparticles at 2.4 wt%, where the sensitivity increased with increasing operating temperature and doping ratios for three temperatures (50, 100, 150) °C. For the pure SnO₂ sample, the sensitivity increased from 1.8% and then increased upon adding Cu nanoparticles to 105% at 150 °C, which is the highest degree reported previously.

Keywords: Nanocomposites; Gas sensing; Tin dioxide; Copper dopants

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1. Introduction

The rising population, industrialization, vehicles, air conditioning units, and other sources contribute to the emission of several gases, adversely impacting the environment. Investigations are underway to identify and alleviate harmful gasses. Emissions from fossil fuel combustion, automotive emissions from internal combustion, and explosions have prompted the advancement of innovative sensing techniques [1-4]. Semiconductor materials are extensively studied in gas sensors owing to their durability and superior performance. In gas-sensing applications, the structure of semiconductor materials and surface modification are crucial. Numerous creative methods leveraging the optical and optoelectronic characteristics of these materials are achieving broad popularity [5,6]. The "solid-gas" interactions transpire on the surfaces of metal oxide semiconductors (MOS) at elevated temperatures ranging from 100 to 300 °C. This temperature is essential to enhance the concentration of free charge carriers, activate surface chemical processes, and facilitate desorption of reaction products. The requirement for heating significantly elevates the power consumption of semiconductor sensors, hence constraining their application in standalone and portable devices, particularly in communication systems such as smartphones. Upon the detection of oxidizing gas (NO₂), sensor signal production transpires due to the adsorption of analyte molecules on the metal oxide surface, resulting in the

localization of electrons on the adsorbed species, which diminishes the electrical conductivity (for n-type semiconductors) [7-14]. Tin dioxide (SnO₂), a globally recognized n-type multifunctional semiconductor, possesses a broad straight bandgap energy of 3.6 eV at 300 K. Moreover, due to the substantial length-to-diameter and surface-to-volume ratios at the nanoscale, the gas sensing and optoelectronic characteristics of one-dimensional SnO₂ and nanostructures exhibit heightened sensitivity to adsorbed species on their surfaces [15]. SnO₂:Cu nanocomposites have elevated electron conductivity and mobility, demonstrating significant applications in transparent electrodes for solar cells [16], anode materials [17], and sensors [18]. The amalgamation of metal (Cu) with tin dioxide (SnO₂) presents a viable strategy for augmenting the physical and chemical attributes of nanocomposites [19]. Copper nanoparticles are highly desirable because to their substantial natural abundance, affordability, and superior catalytic, optical, electrical, and mechanical properties [20]. The gas sensor used in this study was manufactured from SnO₂ metal oxide and doped with (2 and 4%) copper nanoparticles using spray pyrolysis (CSP). We used a comb-type aluminum metal contact for the tool. The optical, structural and microscopic properties of the sample were analyzed to obtain additional information regarding its shape, purity and other properties. Gas sensing measurements were performed, and their selectivity and sensitivity along with different response and recovery rates were

evaluated.

2. Experimental Part

Initially, two types of solutions were prepared, the first solution was tin chloride ($\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$) was used, its molecular weight (225.65 g/mol) and prepared by with a 99% purity. The solution was prepared at a molar concentration of 0.02 M. A 0.4513 g of tin $\text{SnCl}_2 \cdot 2\text{H}_2\text{O}$ was dissolved in 100 ml of deionized water, then the solution was mixed using a magnetic stirrer for 30 min and after completing the dissolution process, we filtered the solution Using special filter papers to make the solution ready for spraying. As for the second solution, the copper was prepared. To prepare it, copper chloride ($\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$) with a molecular weight of 170.48 g/mol and prepared by Alpha Chemika Company with 99% purity was used. The solution was prepared at a molar concentration of 0.02 mol/L. A 0.34096 g of $\text{CuCl}_2 \cdot 2\text{H}_2\text{O}$ was dissolved in 100 ml of deionized water and then added in different percentage volume ratios (0.02 and 0.04) to the aqueous tin dioxide chloride solution.

To fabricate the gas sensor shown in Fig. (1), aluminum (Al) electrodes were coated on the surface of the thin nanostructures produced using the CVD technique. The pressure in the deposition chamber was reduced to 0.1 mbar, and the Al electrodes were evaporated using a comb-shaped multifinger mask to create the metal-semiconductor-metal (MSM) sensor.

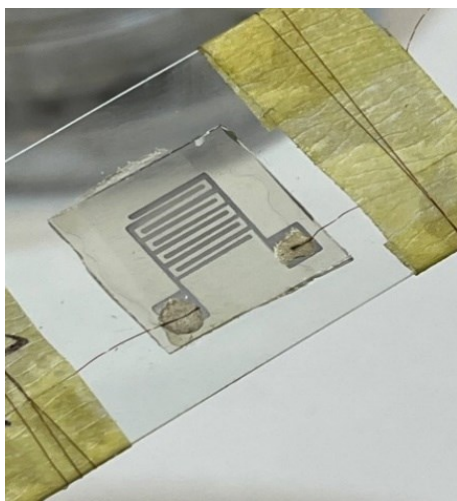


Fig. (1) Photograph of a $\text{SnO}_2\text{:Cu}$ nanocomposite gas sensor device fabricated in this work

3. Results and Discussion

The phase purity and crystalline structure of the produced nanomaterial were examined using x-ray diffraction (XRD). This verifies that the SnO_2 structure is tetragonal. These correspond with the standard JCPDS card no. 01-077-0452. In Fig. (2), the presence of several peaks is observed in the XRD pattern belonging to pure SnO_2 and doped with Cu (2 and 4 %),

at $2\theta = 26.31^\circ$, 33.60° , 37.77° , 51.67° , and 64.23° , corresponding respectively to the crystal planes of (110), (101), (111), (211) and (112), with preferred orientation of plane (110).

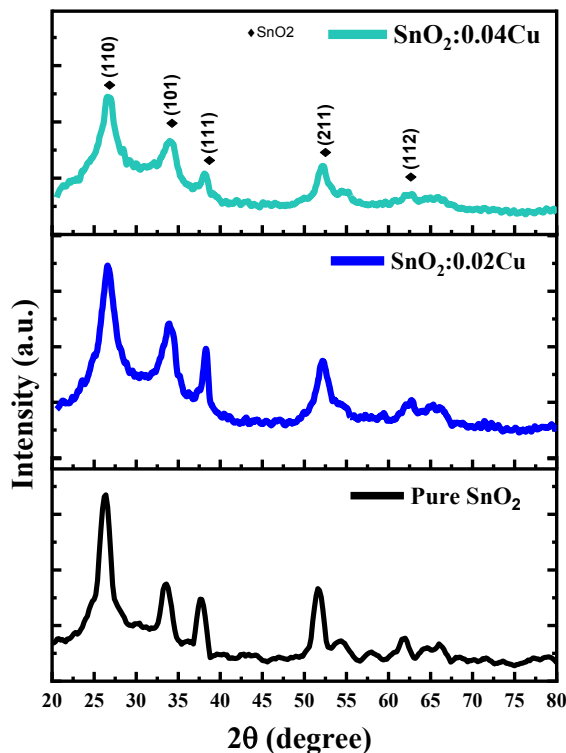


Fig. (2) XRD patterns of (a) pure SnO_2 , (b) $\text{SnO}_2\text{:0.02 Cu}$ and (c) $\text{SnO}_2\text{:0.04 Cu}$

The crystallite size (D) of the pure SnO_2 and doped with Cu nanoparticles was calculated using Debye-Scherrer's equation, expressed by [21]:

$$D = \frac{0.9\lambda}{\beta \cos \theta} \quad (1)$$

where β is the full-width at half maximum (FWHM) of the peak and θ is the Bragg's angle

As can be seen from table (1), there is a variation in the crystallite size when copper ions (Cu^{2+}) are added to the SnO_2 lattice, as they can replace tin atoms (Sn^{4+}) in the crystalline structure. This substitution leads to changes in the crystalline structure such as increasing the regularity of the crystal due to the difference in ionic size between Cu^{2+} and Sn^{4+} . These changes can improve the crystal growth and reduce the structural defects, resulting in an increase in the average size [22].

In order to analyze the surface morphology of $\text{SnO}_2\text{:Cu}$ thin films, field-emission scanning electron microscopy (FE-SEM) was used. Figure (3a) shows that the particles are well connected to each other. The surface of the film is heterogeneous and porous, which makes it more sensitive to gases. The average particle size of the pure film was determined to be 76.44 nm. The SnO_2 doped with 2% and 4% Cu nanoparticles, as shown in figures (3b) and (3c), shows that increasing the doping ratio changes the shape and size of the

particles to a hemispherical shape.

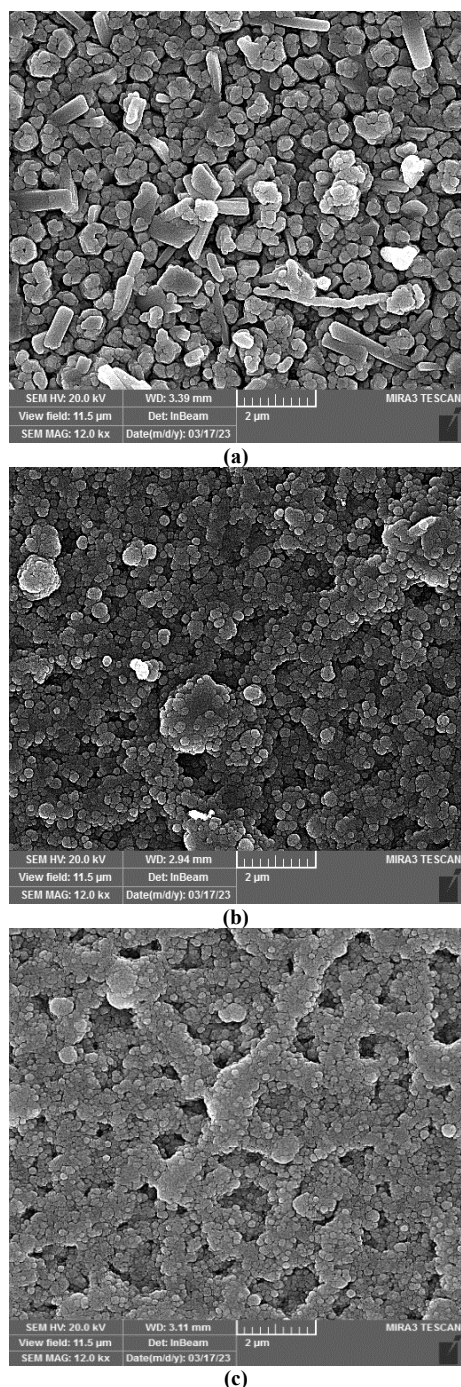


Fig. (3) FE-SEM images of (a) pure SnO₂, (b) SnO₂:0.02 Cu and (c) SnO₂:0.04 Cu

The particles are connected to each other to form aggregated clusters, which are further connected to larger particles randomly distributed across the film surface. The particle size decreases, reaching 38.47 nm at 2% and 25.35 nm at 4%. It is noted that as the percentage of inoculation increases, the shape and size of the granules change to become semi-spherical, as the granules are linked to each other to form clusters of clusters, and these clusters are linked to other large

granules, which are randomly distributed on the surface of the film. The surface texture is rough for the 4% sample, as each crack-like structure consists of smaller nanoparticles or crystalline faces.

The UV-visible absorption spectra of the prepared Cu-doped SnO₂ thin film nanocomposites at 0, 2 and 4% doping ratios are illustrated in Fig (4a) in the wavelength range of 190-11000 nm. These spectra indicated a gradual decrease in the absorbance values with increasing wavelength, which is the behavior of semiconducting SnO₂ films. The absorbance diminished as the concentration of Cu dopant in the SnO₂ lattice increased. The absorption of light by Sn⁴⁺ ions was inhibited by the presence of Cu²⁺ impurity centers. The presence of Cu²⁺ induced defects in the crystal lattice of the storage material, consequently redistributing the atomic orbitals. The abundance of free electrons, characterized by high mobility, diminished the amount of electrons at the ground state, thereby leading to a reduction in the absorbance of materials [23,24].

The bandgap energy (E_g) of the prepared samples (pure SnO₂, SnO₂:0.02 Cu and SnO₂:0.04 Cu) was determined using Tauc's model:

$$(ah\nu)^n = A(h\nu - E_g) \quad (2)$$

where α , $h\nu$, E_g , n , and A represent the absorption coefficient, incident photon energy, energy gap, the exponent n (which varies based on the type of energy transition, e.g., $n=2$ for an indirect bandgap), and the proportionality constant, respectively [25]

In Fig. (4b), the synthesized pure SnO₂ exhibits a band gap of 3.021 eV, which increases with higher concentrations of copper, reaching 3.241 eV for the SnO₂:0.02 Cu sample and 3.283 eV for the SnO₂:0.04 Cu sample. This notable increase in band gap with Cu nanoparticles concentration is attributed to the quantum confinement effect. When ratio Cu increases, below band absorption transpires. As the concentration of Cu rises, the Urbach energy correspondingly increases, indicating a rise in distortion. The rise in distortion may result from two factors; either copper is generating levels adjacent to the conduction band, or the inclusion of copper into the system is creating oxygen vacancies that produce levels near the conduction band [26].

Gas sensors for NO₂ gas based on nanocomposites SnO₂:Cu have been fabricated, was examined at three temperatures (50, 100, and 150°C) at a constant flow rate of NO₂ gas by measuring the resistance (R M Ω) over time across two multi-finger electrodes. The transient response of the NO₂ gas sensor utilizing SnO₂:Cu is illustrated in Fig. (5). Three cycles were consecutively documented throughout measurement duration of 100 s. These cycles were documented in relation to various ultra-three temperatures of NO₂ gas. The reaction of SnO₂ to NO₂ gas was significantly influenced by surface shape. The sensor response ($S\%$) is calculated by equation [27]:

$$S\% = \frac{R_a - R_g}{R_a} \times 100\% \quad (3)$$

where R_a represents resistance in the air and R_g represents the resistance when gas is released

The resistance of the SnO_2 -based sensor escalated following exposure to NO_2 gas, with a rapid increase in response until maximum values were attained, as illustrated in Figures. Upon substituting NO_2 gas with air, the sensor resistance reverted to its original state, signifying a reversible adsorption of NO_2 molecules on the SnO_2 surface. The sensor's response to NO_2 gas exhibits a distinct increase as the temperature rises from 50 to 150 °C.

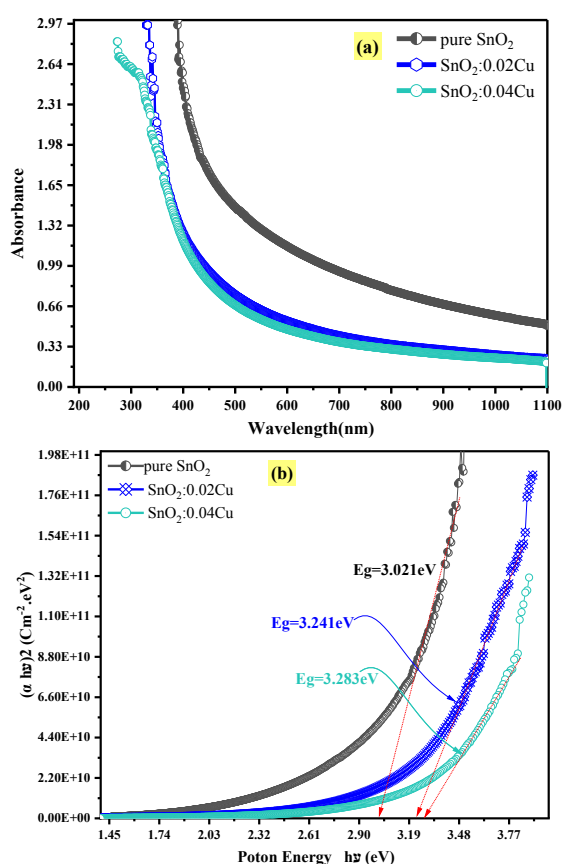


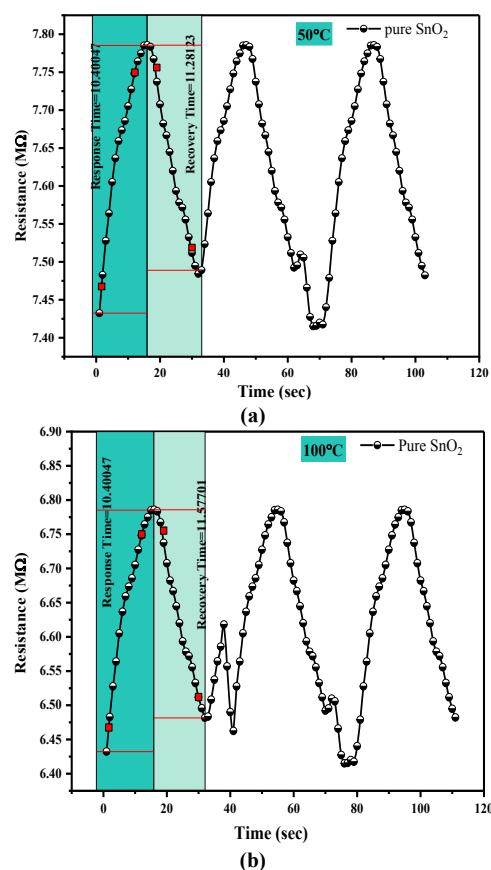
Fig. (4) (a) UV-visible absorption spectra and (b) energy gap plots of pure SnO_2 , $\text{SnO}_2:0.02 \text{ Cu}$ and $\text{SnO}_2:0.04 \text{ Cu}$

The sensitivity of NO_2 gas at working temperatures. The dynamic sensing transients of sensor upon exposure of NO_2 gas. When the operating temperature increased, there was an increase in sensitivity within the range of $S=4.1$ to 8% for the three degrees reported above and referred to in table (2). As the temperature rises, the activity of the adsorbed gas molecules intensifies, leading to heightened interactions that amplify the variation in the film's electrical resistance. The acceleration of chemical processes enhances the efficiency of the contact between the poisonous gas and the sensitive surface, resulting in a significant alteration in electrical resistance, hence increasing sensitivity

[28,29]. An improvement in the response of the SnO_2 -based gas sensor was observed when the structure was enhanced by adding copper nanoparticles at percentages of 2 and 4% by weight. The sensitivity of the $\text{SnO}_2:0.02 \text{ Cu}$ film when gas opened was about 6.7% at 50 °C, and it increased to a value of 10.4% at 150 °C. While the improvement was noted for the $\text{SnO}_2:0.04 \text{ Cu}$ film over its predecessor, from ~11.4% to 105% at 150°C, with an improvement in response and recovery times.

Table (2) Response time, recovery time and gas sensitivity of gas sensors at operating temperatures of 50, 100 and 150 °C

Samples	T (°C)	Response Time (s)	Recovery Time (s)	Sensitivity (%)
Pure SnO_2	50	10.40	11.28	4.1
	100	10.40	11.57	4.7
	150	9.66	16.19	8.1
$\text{SnO}_2:0.02 \text{ Cu}$	50	6.69	18.16	6.7
	100	9.75	14.64	7.9
	150	8.09	13.72	10.4
$\text{SnO}_2:0.04 \text{ Cu}$	50	12.52	17.50	11.4
	100	10.41	17.81	9.6
	150	5.84	16.00	105



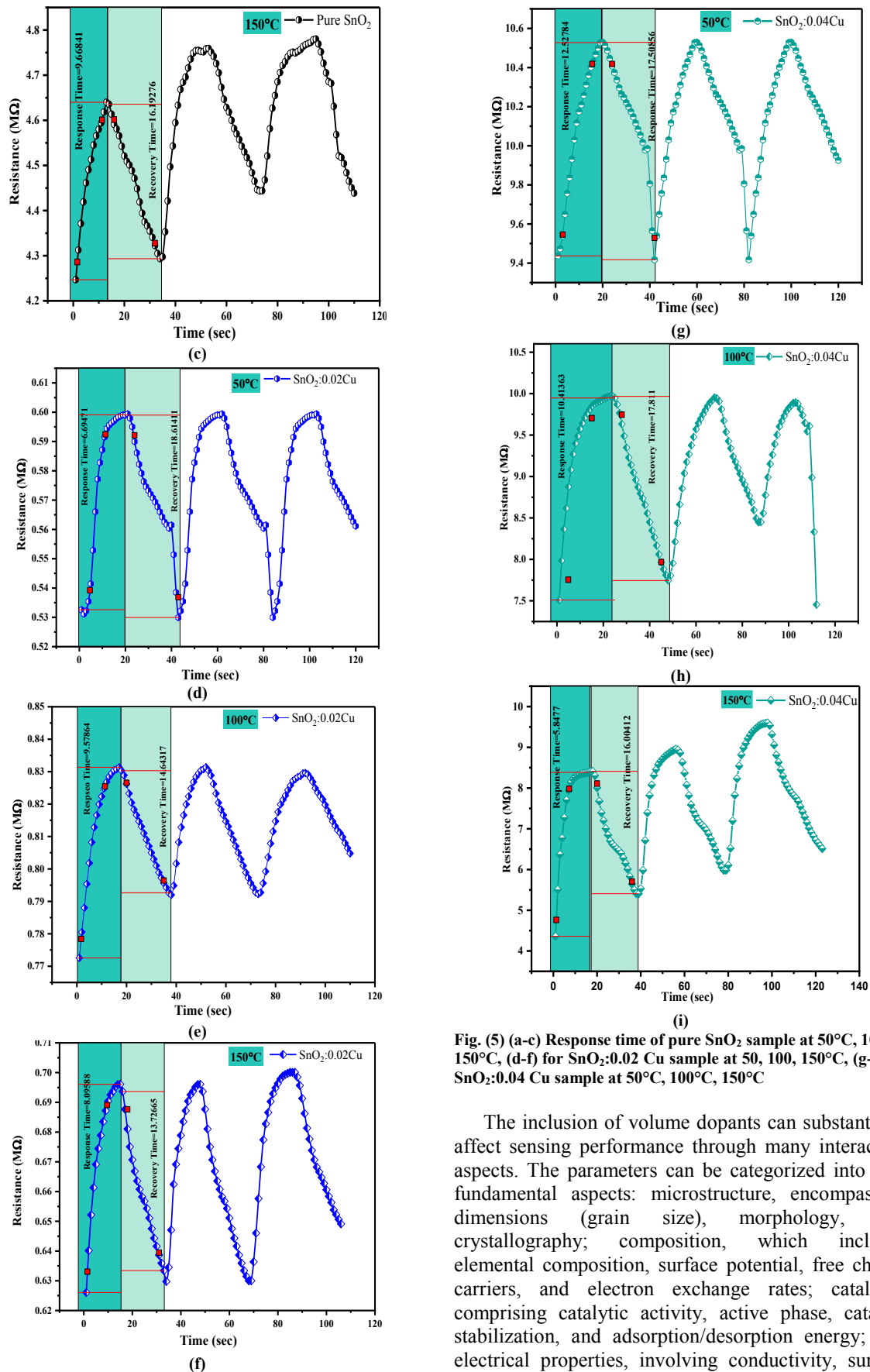


Fig. (5) (a-c) Response time of pure SnO₂ sample at 50°C, 100°C, 150°C, (d-f) for SnO₂:0.02 Cu sample at 50, 100, 150°C, (g-i) for SnO₂:0.04 Cu sample at 50°C, 100°C, 150°C

The inclusion of volume dopants can substantially affect sensing performance through many interaction aspects. The parameters can be categorized into four fundamental aspects: microstructure, encompassing dimensions (grain size), morphology, and crystallography; composition, which includes elemental composition, surface potential, free charge carriers, and electron exchange rates; catalysis, comprising catalytic activity, active phase, catalyst stabilization, and adsorption/desorption energy; and electrical properties, involving conductivity, surface

potential, free charge carriers, and electron exchange rates. The dopants can boost sensor qualities through the production and stabilization of smaller grains, the increase of nanostructure porosity, and the improvement of long-term stability [30,31,33]. Nonetheless, the development of an optimal sensor material continues to be problematic, as the enhancement of a beneficial attribute may concurrently impair another or exacerbate a detrimental characteristic.

4. Conclusion

The incorporation of copper nanoparticles into the tin oxide (SnO_2) structure showed an improvement in the structural, optical and gas detection properties of SnO_2 films deposited using the chemical spray deposition technique. The incorporation of copper at 2% or 4% enhanced the crystallinity of the films. The absorbance value decreased, while the energy gap increased from 3.02 to 3.28 eV. The fabricated NO_2 gas sensors showed a significant improvement at ratio SnO_2 -0.04 Cu, achieving the highest response of 105% at an operating temperature of 150°C.

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Table (1) Some structural parameters from XRD data

Sample	2θ (deg)	d(Å)	Intensity	FWHM (deg)	(hkl)	Crystallin size (nm)	Average Crystallin size (nm)
SnO ₂ pure	26.31	3.38	184.5555	0.5904	110	13.09	10.79
	33.60	2.66	65.84702	0.984	101	7.72	
	37.77	2.38	127.4064	0.7872	200	9.54	
	51.67	1.76	107.7891	0.5904	211	12.10	
	61.92	1.49	33.47593	0.5904	310	11.53	
SnO ₂ :0.02Cu	26.49	3.36	839.2017	0.393	110	19.63	14.78
	33.47	2.60	339.935	0.492	210	15.45	
	37.182	2.35	197.4313	0.7872	200	9.561	
	52.334	1.74	342.6136	0.492	211	14.48	
SnO ₂ :0.04Cu	26.54	3.35	907.1252	0.688	110	11.23	14.67
	33.801	2.59	258.9784	0.393	210	19.33	
	38.418	2.34	461.0662	0.3936	200	19.05	
	51.72	1.75	361.0776	0.787	211	9.08	