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Effect of Precursor Concentration on Structural Characteristics of Titanium Dioxide Nanopowders Synthesized by Green Method

In this work, titanium dioxide nanopowders were synthesized by an eco-friendly green method. The structural characteristics of the synthesized nanopowders were determined, analyzed and compared to introduce their dependency on the precursor concentration. It was found that the synthesized nanopowders show polycrystalline structures with a reasonable dependency on the concentration of the precursor of the TiO₂ (i.e., titanium isopropoxide). Also, the minimum particle size was found to increase with increasing the concentration of the precursor. Higher concentration of the precursor results in much more stoichiometric compound. The lowest concentration (0.5 mL) has resulted in higher surface roughness. The synthesized nanopowders will be employed in biological and agricultural applications to make use of the photocatalytic activity of TiO₂.

Keywords: Titanium dioxide; Nanopowders; Green synthesis, Reduction method **Received:** 20 March 2025; **Revised:** 28 April 2025; **Accepted:** 04 May 2025

1. Introduction

Eco-friendly nanomaterial manufacturing methods, known as green technologies, offer numerous advantages compared to other methods and techniques. They are characterized by low cost, minimal requirements, high availability, and rapid production. They also can be applied near the field, saving time, effort, storage, and transportation [1-4].

One of the most commonly used green methods for producing nanomaterials is the chemical reduction of nanomaterials from their chemical precursors using plant extracts [5]. This method can produce a variety of nanomaterials depending on the available plant-based components in the surrounding environment [6]. It offers several advantages that other methods cannot provide, primarily the absence of harmful emissions and the ability to utilize plant residues by converting them into materials that can be easily returned to the soil as stimulants and enhancers [7,8]. Also, it does not require complex or costly procedures and ensures the abundant production of nanomaterials, making it a viable source for applications that demand relatively large quantities of the desired nanomaterials [9-11].

The interest in using nanomaterials in biological and agricultural applications is continuously increasing due to the unique properties and characteristics these materials exhibit, which cannot be offered by conventional chemical fertilizers and growth stimulants [12-14]. Titanium dioxide stands out among these materials, especially when prepared in the form of a nanopowder, as it benefits from the synergistic effect of the quantum size of the nanoparticles and the photocatalytic activity that this material possesses [15,16].

Titanium dioxide nanoparticles (TiO₂ NPs) have gained increasing attention in agricultural research due

their nanoscale dimensions and favorable physicochemical properties [17-20]. These include a high surface area-to-volume ratio, strong ultraviolet (UV) absorption, high photocatalytic activity, and exceptional chemical stability [21-24]. These features make TiO₂ NPs highly suitable for enhancing plant physiological functions and protecting crops from various biotic and abiotic stresses [25]. The agricultural behavior and efficacy of TiO2 NPs are determined by properties such as their crystalline phase (anatase vs. rutile), particle size, surface charge, and tendency to aggregate. The anatase phase, in particular, exhibits superior photocatalytic activity, enabling more efficient interaction with light and promotion of photosynthesis [26-28]. However, it can present phytotoxicity risks at elevated concentrations [25]. TiO₂ NPs may interact with plant surfaces through foliar application or enter the root zone via the soil, where environmental factors like pH, moisture, and microbial communities influence their mobility and bioavailability [29,30]. One of the most recognized benefits of TiO₂ NPs in agriculture is their ability to enhance photosynthetic performance. These nanoparticles have been shown to promote chlorophyll biosynthesis, optimize light absorption, and stimulate photosynthetic enzyme activity [31]. Their lightscattering and UV-blocking properties help reduce photodamage to chloroplasts and improve photon use efficiency [25]. For instance, in Vetiveria zizanioides (vetiver grass), TiO₂ NPs significantly enhanced physiological parameters such as net photosynthetic rate, stomatal conductance, and transpiration rate [32]. TiO₂ NPs are known to exhibit hormetic effects wherein low concentrations stimulate plant growth and metabolic functions, while higher doses induce inhibitory or toxic effects. At optimal doses, they



promote root elongation, shoot growth, and resilience to environmental stresses. In contrast, excessive exposure may lead to oxidative stress, chlorosis, and disruption of DNA integrity in plant cells [25]. This biphasic response underlines the importance of dosage optimization for the safe and beneficial use of TiO₂ NPs in agricultural applications [33,34].

During the last three decades, several methods and techniques, such as sol-gel, pulsed-laser deposition, chemical bath deposition, reactive sputtering, solvothermal and chemical reduction, were used to prepare TiO₂ nanoparticles with highly controlled characteristics to serve certain applications [35-38]. Each method or technique may have advantages over the other ones, however, the solvothermal chemical method was selected in this work due to the advantages mentioned before [39,41].

Titanium dioxide nanopowders were prepared by the chemical reduction of titanium isopropoxide as a precursor using aqueous solution of banana peels. The structural characteristics of the prepared nanopowders were determined and analyzed to introduce the effect of precursor concentration on these characteristics.

2. Experimental Work

Figure (1) shows schematically the experimental setup used for the preparation of TiO₂ nanopowders. Banana peels were used as a resource for plant-derived titanium, while titanium isopropoxide (C₁₂H₂₈O₄Ti) served as the precursor material. Fresh, unripe bananas were obtained from the local market, and their peels were cut into small pieces, washed three times with deionized distilled water (DDW) to remove any impurities, and dried using absorbent paper. Then, 80 g of the dried peels were weighed and placed in a beaker containing 150 mL of DDW. The mixture was heated up to its boiling point (100°C) and maintained at this point for 20 minutes. After boiling, the mixture was filtered using Whatman® #1 filter paper to remove solid residues.

In a separate flask, an aqueous solution was prepared by dissolving certain volume (0.5, 1, and 2 mL) of C₁₂H₂₈O₄Ti in 10 mL of DDW. This solution was heated up to 40°C on a hotplate stirrer for 10 minutes with continuous stirring to ensure complete dissolution and homogeneity. The heating step was then turned off, but stirring was continued while immediately preceding to the next step.

The banana peel extract was gradually added to the $C_{12}H_{28}O_4Ti$ solution in 5 mL increments while stirring continuously for one hour to promote nanoparticle formation. The chemical reaction of the precursor with the aqueous solution to produce titanium dioxide as a solid nanopowder and isopropyl alcohol is shown as

 $C_{12}H_{28}O_4Ti(1) + 2H_2O(1) \rightarrow TiO_2(s) + 4(CH_3)_2CHOH(1)$

The resulting mixture was then filtered using Whatman® #1 filter paper to collect the synthesized nanopowder. This nanopowder was thoroughly washed with DDW to remove any remaining impurities from the previous steps. Subsequently, the collected nanopowder was subjected to controlled heating until it turned dark grey. Finally, the material underwent a potassium hydroxide (KOH) treatment to enhance purity and eliminate any residual organic matter.

The synthesized samples were characterized by x-ray diffraction (XRD) using PanAnalytical ARIES x-ray diffractometer, field-emission scanning electron microscopy (FE-SEM), energy-dispersive x-ray spectroscopy (EDS) using an FEI InspectTM 50 FE-SEM instrument, and nanosurf atomic force microscopy (AFM).



Fig. (1) Schematic diagram of the experimental setup used in this work

3. Results and Discussion

Figure (1) shows the XRD patterns of the synthesized nanopowder samples using different concentrations of the precursor (C₁₂H₂₈O₄Ti). For lower concentrations (0.5 and 1 mL), the prepared TiO₂ material contains anatase phase only, while the higher concentration (2 mL) has resulted in the appearance of four diffraction peaks belonging to the rutile phase. This is attributed to the increase in enthalpy of the reaction with increasing the concentration of precursor, which lead to induce the phase transformation from anatase to rutile. Despite that the transformation temperature is determined at 600°C for bulk structures of TiO₂, at the nanoscale, reasonably lower temperatures may induce this phase



transformation as the surface-to-volume of the prepared material is increased by three orders of magnitude when the dimensions are reduced from micro to nanoscale [42].

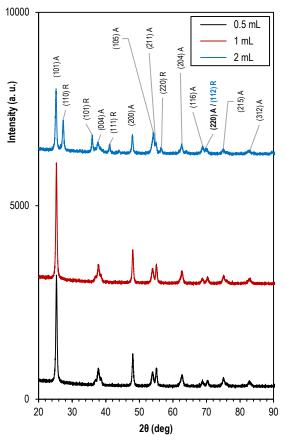
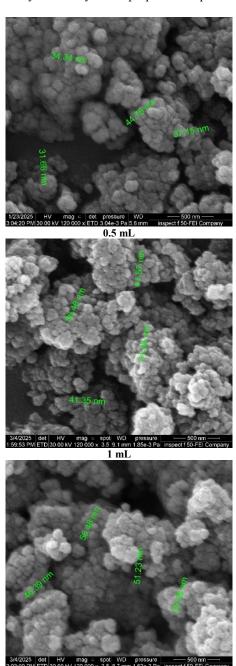


Fig. (1) XRD patterns of the synthesized nanopowders using different concentrations of the $C_{12}H_{28}O_4Ti$ precursor

Table (1) shows the crystalline structural parameters of the prepared samples obtained from XRD results. It is clear that the average crystallite size increases with increasing the concertation of the precursor with a featured appearance of the rutile distinct peak at 27.28°. Another important finding related to the increase of precursor concentration is the observed decrease in the microstrain of the prepared sample. Such decrease may result in a decrease in light absorption by the prepared TiO₂ sample [43]. Therefore, the structural characteristics of the prepared nanomaterial should be carefully analyzed and controlled due to the correlation to the spectroscopic characteristics, which are required to be optimized in applications based on photocatalytic activity of the prepared material.

Figure (2) shows the FE-SEM images of the synthesized nanopowder samples using different concentrations of the precursor (C₁₂H₂₈O₄Ti). For all concentrations, the prepared samples show spherical particles with reasonable agglomerated regions. However, the minimum particle size was found to

increase with increasing the concentration of the precursor (Fig. 3). This is attributed to the role of higher concentration to produce much more ${\rm TiO_2}$ nanoparticles during the same period time taken by the reduction reaction to complete. Therefore, the probability of the grains to grow and form larger particles is increased. The increase in the minimum particle size is about 12% as the precursor concertation is increased from 0.5 to 2 mL, and this value is not sufficiently effective in changing the photocatalytic activity of the prepared sample.



2 mL Fig. (2) FE-SEM images of the synthesized nanopowders using different concentrations of the C₁₂H₂₈O₄Ti precursor



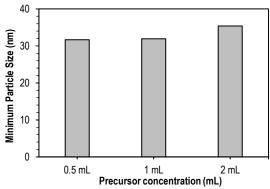


Fig. (3) Variation of minimum particle size of the nanopowder samples with concentration of the $C_{12}H_{28}O_4Ti$ precursor

It was confirmed that the stoichiometry of the TiO₂ materials may affect their spectroscopic characteristics, which are very sensitive to the elemental structure of the material. Therefore, the elemental analysis of the prepared samples is presented in EDS spectra shown in Fig. (4). All samples showed high structural purity as no peaks belonging to elements other than titanium, oxygen and carbon were found. Carbon is originated from two sources, the residual isopropyl alcohol (which is a result of the reduction reaction) and the mounting method used in EDS test.

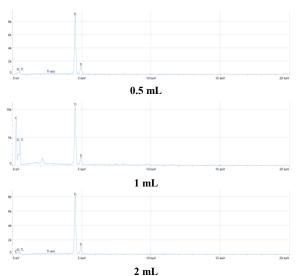


Fig. (4) EDS spectra of the synthesized nanopowders using different concentrations of the $C_{12}H_{28}O_4Ti$ precursor

Figure (5) indicates the variation of the elemental contents in the prepared samples with the precursor concentration. The ratio of [O]/[Ti] represents the indicator of the sample stoichiometry and the sample prepared using the highest concentration (2 mL) showed a [O]/[Ti] ratio of 1.9, which is very close to the typical value of 2. This result is ascribed to the role of higher concentration in providing more

titanium atoms for the reaction to form titanium dioxide molecules.

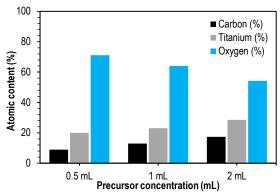


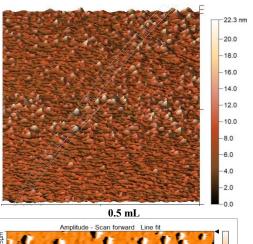
Fig. (5) Variation of atomic contents of carbon, titanium and oxygen in the nanopowder samples with concentration of the $C_{12}H_{28}O_4Ti$ precursor

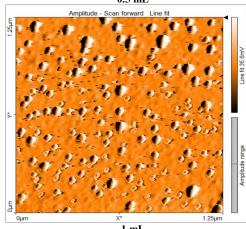
Figure (6) shows the 2D AFM images of the synthesized nanopowder samples using different concentrations of the precursor (C₁₂H₂₈O₄Ti). The surface roughness of the prepared material is very advantageous for the applications based on the radiation-matter interaction as the high surface roughness provide larger surface for such interaction and hence higher opportunity for the aimed events. Apparently, the lowest concentration (0.5 mL) has resulted in higher surface roughness due to the lowest particle size of the nanostructures within this sample.

The surface roughness of a nanomaterial can significantly influence its photocatalytic activity as a rougher surface provides a higher surface area, offering more active sites for photocatalytic reactions. This enhances the interaction between the catalyst, reactants, and light, leading to improved reaction rates. Also, surface roughness can enhance light trapping through increased light scattering and reflection. This allows for more effective absorption of light, particularly in materials like titanium dioxide (TiO₂) that rely on UV or visible light for photocatalysis. Additionally, a rough surface often introduces more surface defects or irregularities that can act as charge carrier traps. While some defects may facilitate charge separation and prevent recombination, excessive surface defects might lead to charge recombination, reducing photocatalytic efficiency. Nanomaterials with rough surfaces typically exhibit improved mass transfer of reactants and products. The porous or uneven texture can facilitate better diffusion, accelerating reaction rates. Finally, surface roughness can increase hydrophilicity of nanomaterials, enhancing adsorption of water and pollutants. Improved adsorption effective promotes photocatalytic degradation of contaminants. Consequently, optimizing the surface roughness to balance increased



surface area and minimized charge recombination is crucial for maximizing photocatalytic performance.





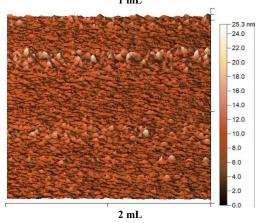


Fig. (6) FE-SEM images of the synthesized nanopowders using different concentrations of the $C_{12}H_{28}O_4Ti$ precursor

4. Conclusion

In concluding remarks, TiO_2 nanopowders were successfully synthesized by an eco-friendly green synthesis method. The synthesized nanopowders show polycrystalline structures with a reasonable dependency on the concentration of the precursor of the TiO_2 (i.e., titanium isopropoxide). As this concentration is increased, the rutile phase of TiO_2 has appeared much more apparently than lower

concentrations. Also, the minimum particle size was found to increase with increasing the concentration of the precursor due to the role of higher concentration to produce much more TiO₂ nanoparticles during the same period time taken by the reduction reaction to complete. Higher concentration of the precursor results in much more stoichiometric compound due to the role of higher concentration in providing more titanium atoms for the reaction to form titanium dioxide molecules. Finally, the lowest concentration (0.5 mL) has resulted in higher surface roughness due to the lowest particle size of the nanostructures within this sample. The synthesized nanopowders will be employed in biological and agricultural applications to make use of the photocatalytic activity of TiO₂.

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Table (1) Crystalline structural parameters obtained from XRD results

0.5 mL				
2θ (deg)	d-spacing (Å)	FWHM Left (deg)	Crystallite Size (nm)	Micro Strain (%)
25.2834	3.51970	0.4187	17.8	0.98841
37.7723	2.37975	0.6393	9.2	1.29228
38.5727	2.33220	0.3876	17.1	0.68038
48.0155	1.89328	0.3985	19.5	0.48580
53.8925	1.69986	0.6539	11.7	0.72620
55.0374	1.66718	0.4336	17.6	0.47451
62.6063	1.48258	0.8014	9.3	0.80082
68.7625	1.36409	0.6569	10.9	0.62375
70.2629	1.33859	0.4972	16.3	0.41044
75.0628	1.26445	0.7445	8.9	0.71267
82.7230	1.16569	1.0303	10.8	0.54155
			13.55	0.70334
1 mL				
			A	111 04 1 (0/)
2θ (deg)	d-spacing (Å)	FWHM Left (deg)	Crystallite Size (nm)	Micro Strain (%)
2θ (deg) 25.2735	d-spacing (A) 3.52106	FWHM Left (deg) 0.4172	Crystallite Size (nm)	0.97116
		, 0,	, ,	, ,
25.2735	3.52106	0.4172	18.1	0.97116
25.2735 36.9304	3.52106 2.43205	0.4172 0.6559	18.1	0.97116 0.88019
25.2735 36.9304 37.7917	3.52106 2.43205 2.37858	0.4172 0.6559 0.6289	18.1 13.8 13.5	0.97116 0.88019 0.88357
25.2735 36.9304 37.7917 38.5615	3.52106 2.43205 2.37858 2.33285	0.4172 0.6559 0.6289 0.5462	18.1 13.8 13.5 16.6	0.97116 0.88019 0.88357 0.70301
25.2735 36.9304 37.7917 38.5615 48.0131	3.52106 2.43205 2.37858 2.33285 1.89337	0.4172 0.6559 0.6289 0.5462 0.4173	18.1 13.8 13.5 16.6 19.1	0.97116 0.88019 0.88357 0.70301 0.49505
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25.2735 36.9304 37.7917 38.5615 48.0131 53.8941 55.0362 62.6035	3.52106 2.43205 2.37858 2.33285 1.89337 1.69981 1.66721 1.48264	0.4172 0.6559 0.6289 0.5462 0.4173 0.6440 0.4367 0.7935	18.1 13.8 13.5 16.6 19.1 12.1 17.5 9.6	0.97116 0.88019 0.88357 0.70301 0.49505 0.70371 0.47575 0.76978
25.2735 36.9304 37.7917 38.5615 48.0131 53.8941 55.0362 62.6035 68.7263	3.52106 2.43205 2.37858 2.33285 1.89337 1.69981 1.66721 1.48264 1.36472	0.4172 0.6559 0.6289 0.5462 0.4173 0.6440 0.4367 0.7935 0.6532	18.1 13.8 13.5 16.6 19.1 12.1 17.5 9.6	0.97116 0.88019 0.88357 0.70301 0.49505 0.70371 0.47575 0.76978 0.57628
25.2735 36.9304 37.7917 38.5615 48.0131 53.8941 55.0362 62.6035 68.7263 70.2667	3.52106 2.43205 2.37858 2.33285 1.89337 1.69981 1.66721 1.48264 1.36472 1.33853	0.4172 0.6559 0.6289 0.5462 0.4173 0.6440 0.4367 0.7935 0.6532 0.5550	18.1 13.8 13.5 16.6 19.1 12.1 17.5 9.6 11.8	0.97116 0.88019 0.88357 0.70301 0.49505 0.70371 0.47575 0.76978 0.57628 0.45988
25.2735 36.9304 37.7917 38.5615 48.0131 53.8941 55.0362 62.6035 68.7263 70.2667 75.0738	3.52106 2.43205 2.37858 2.33285 1.89337 1.69981 1.66721 1.48264 1.36472 1.33853	0.4172 0.6559 0.6289 0.5462 0.4173 0.6440 0.4367 0.7935 0.6532 0.5550 0.7048	18.1 13.8 13.5 16.6 19.1 12.1 17.5 9.6 11.8 14.6	0.97116 0.88019 0.88357 0.70301 0.49505 0.70371 0.47575 0.76978 0.57628 0.45988 0.69345



2 mL				
2θ (deg)	d-spacing (Å)	FWHM Left (deg)	Crystallite Size (nm)	Micro Strain (%)
25.1466	3.53854	0.4283	17.7	0.99734
27.2800	3.26646	0.4406	15.9	1.02667
35.9330	2.49723	0.3456	22.8	0.54686
37.6731	2.38579	0.8776	5.4	2.20809
41.1075	2.19405	0.3508	21.7	0.50491
43.8429	2.06329	0.5673	13.0	0.79545
47.8876	1.89803	0.3959	18.9	0.50103
54.1098	1.69354	0.8016	8.9	0.95158
54.9514	1.66958	0.3800	24.7	0.33733
56.4886	1.62774	0.4395	17.6	0.46269
62.5461	1.48386	0.6873	9.9	0.75112
63.9862	1.45390	0.3490	23.4	0.31087
68.7909	1.36360	0.6599	11.0	0.62222
69.9537	1.34375	0.8443	12.0	0.55902
74.9809	1.26563	0.9734	7.2	0.87294
82.5061	1.16821	1.1333	9.8	0.59823
89.4831	1.09431	0.4025	30.5	0.17966
			15.90	0.55879